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# **Research Article**

# Hexagonal Lead Ferrite Magnetic Separation Catalysts: Synthesis, Optical Characterization, Ultrasonic Catalytic Activity and Performance Prediction

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# Abstract:

**Objective:** This study was performed to explore a new method to synthesize  $PbFe_{12}O_{19}$  magnetic separation catalysts and analyze its ultrasonic catalytic activity for the degradation of rhodamine B (RhB) dye under ultrasonic vibration.

**Methods:** A polyacrylamide gel method was developed to synthesize the hexagonal lead ferrite ( $PbFe_{12}O_{19}$ ) magnetic separation catalysts with different Fe/Pb molar ratios and sintering temperatures.

**Results:** The phase structure and composition of  $PbFe_{12}O_{19}$  magnetic separation catalysts altered with the change in the Fe/Pb mole ratio and sintering temperature. The color properties, energy band values and degradation percentages of  $PbFe_{12}O_{19}$  magnetic separation catalysts for the degradation of RhB dye under ultrasonic vibration were linearly independent of Fe/Pb mole ratio and sintering temperature. The maximum degradation percentage of 78.99% for the PbFe\_{12}O\_{19} magnetic separation catalysts for the removal of RhB from wastewater was found at 2g/L of initial catalyst concentration, 50mg/L of initial dye concentration, and 180min of vibration time.

**Conclusion:** The  $PbFe_{12}O_{19}$  magnetic separation catalysts exhibited high catalytic activity for the degradation of RhB dye under ultrasonic vibrations. Based on the experiment parameters, a neural network model was established to predict the effect of synthesis parameters on the degradation percentage of  $PbFe_{12}O_{19}$  catalysts. This work further confirmed that the neural network algorithm was available to predict the ultrasonic catalytic activity of semiconductor materials.

**Keywords:** PbFe<sub>12</sub>O<sub>19</sub>, polyacrylamide gel method, neural network model, rhodamine B, degradation percentage

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#### **1 INTRODUCTION**

Catalysts are preferred materials for the degradation of organic dye wastewater due to their high specific surface area, high charge transfer and separation efficiency, and high degradation activity<sup>[1-6]</sup>. Depending on the different degradation methods of organic dyes, various catalysts, including the photocatalysts, thermal catalysts, electrocatalysts, piezoelectric catalysts, ultrasonic catalysts, and adsorbents have been developed to effectively use of various physical and chemical effects to degrade organic dyes<sup>[7-13]</sup>. Among these catalysts, ultrasonic catalyst is an efficient catalyst that uses the energy generated by ultrasonic vibration to promote the transfer and separation of electrons and holes inside the catalyst, which consequently results in oxidation or reduction reaction with dyes<sup>[14,15]</sup>. However, this technology is still rudimentary, so it is of great significance to explore a new ultrasonic catalyst and study its ultrasonic catalytic activity.

In addition to the effective degradation of organic wastewater, catalysts may also cause secondary pollution to water bodies<sup>[16,17]</sup>. To effectively utilize the degraded wastewater, catalysts and water must be efficiently separated<sup>[18]</sup>. Magnetic separation of catalyst and water using magnetic field is an effective way to solve the above issues<sup>[19-21]</sup>. Hexagonal lead ferrite (PbFe<sub>12</sub>O<sub>19</sub>) is a magnetic semiconductor material, which is often used in magnetic devices, magnetic recording medias, catalysts, and other related fields<sup>[22-25]</sup>. Moreover, the ultrasonic catalytic activity of PbFe<sub>12</sub>O<sub>19</sub> has been sporadically. Therefore, using a special preparation method to synthesize PbFe<sub>12</sub>O<sub>19</sub> and study its ultrasonic catalytic activity may provide technical guidance for the field of ultrasonic catalysis.

The polyacrylamide gel method is an effective method for the synthesis of metal oxide semiconductor materials by our research group<sup>[26-29]</sup>. Different morphologies and properties of metal oxide semiconductor materials can be obtained by adjusting and controlling the experimental parameters<sup>[30-32]</sup>. By collecting multiple experimental data, the establishment of reasonable models and prediction of the physical and chemical properties of semiconductor materials constitute a major development trend of material synthesis research. Through predicting the physical and chemical properties of semiconductor materials, the experimental cost can be greatly reduced. Artificial neural network algorithm model shows unique advantages in solving such problems because of its unique predictive function<sup>[33]</sup>. In the current research, the PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts with different Fe/Pb molar ratios and sintering temperatures were prepared by a polyacrylamide gel method. The effects of different the Fe/Pb molar ratios and sintering temperatures on the physicochemical properties of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts were studied systematically. According to the nonlinear relationship between synthetic parameters and experimental results, an appropriate artificial neural network model was established to predict the physicochemical properties of PbFe<sub>12</sub>O<sub>19</sub>.

# 2 MATERIALS AND METHODS 2.1 Materials

Ferrous chloride tetrahydrate, lead acetate trihydrate, citric acid, glucose, acrylamide, N,N'-methylene diacrylamide, and rhodamine B (RhB) were purchased from Chengdu Aike Reagent Co., Ltd. All the above reagents were of analytical grade and were used without further purification.

# 2.2 Synthesize of PbFe<sub>12</sub>O<sub>19</sub> Magnetic Separation Catalysts

Stoichiometric amounts of ferrous chloride tetrahydrate and lead acetate trihydrate were dissolved in the distilled water in turn to obtain a solution of 0.015mol/L with the total cations. To study the effect of Fe/Pb molar ratio on the phase purity of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts, the molar ratios of Fe/ Pb were 2, 3, 4, 5, 6, 9 and 12, respectively. The Pb/ Fe molar ratios of 2, 3, 4, 5, 6, 9 and 12 were labeled as samples S1, S2, S3, S4, S5, S6 and S7, respectively. After the above reagents were completely dissolved, 4.7282g citric acid, 20g glucose, 9.5958g acrylamide, and 1.9192g N,N'-methylene diacrylamide were added to the above solution in turn. The citric acid, glucose and N, N'-methylene diacrylamide were used as chelating agents and prevention of gel collapse during drying and a cross-linking agent. After all reagents were completely dissolved, the solution was heated to 80-100°C to initiate the polymerization of acrylamide and N, N'-methylene diacrylamide to obtain a jelly-like gel. The jelly-like gel was then placed in a drying oven to dry at 120°C for 24h. The final PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts were obtained by sintering the xerogel at different temperatures. The PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts with different Fe/Pb molar ratios including 2, 3, 4, 5, 6, 9 and 12 and calcined at 600, 700, 800, 900, 1000 and 1100°C are labeled as S1-600, S2-600, S3-600, S4-600, S5-600, S6-600, S7-600, S1-700, S2-700, S3-700, S4-700, S5-700, S6-700, S7-700, S1-800, S2-800, S3-800, S4-800, S5-800, S6-800, S7-800, S2-900, S2-1000, S2-1100, S5-900, S5-1000, S5-1100, S1-1100, S31100, S4-1100, S6-1100 and S7-1100, respectively. The preparation flow chart of  $PbFe_{12}O_{19}$  magnetic separation catalysts prepared by the polyacrylamide gel method with different Fe/Pb molar ratios is shown in Figure 1.

#### 2.3 Material Characterization

The crystal structure and phase purity of  $PbFe_{12}O_{19}$ magnetic separation catalysts were recorded using a DX-2007BH X-ray diffractometer (XRD). Fourier transform infrared (FTIR) spectra of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts in the range 400-4000cm<sup>-1</sup> were recorded using a FTIR-650 spectrometer. Ultraviolet-visible (UV-Visible) diffuse reflectance spectra of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts were measured by a UV1800 UV-Visible spectrophotometer. The charge state of  $PbFe_{12}O_{19}$ magnetic separation catalysts was characterized by a KRATOS X SAM 800 X-ray photoelectron spectrometer (XPS). The surface morphology of  $PbFe_{12}O_{19}$  magnetic separation catalysts was characterized by a scanning electron microscopy (SEM) and transmission electron microscopy (TEM). The magnetic properties of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts were measured by a superconducting quantum interference device.

#### 2.4 Ultrasonic Catalytic Experiments

The ultrasonic catalytic experiments of  $PbFe_{12}O_{19}$ magnetic separation catalysts for the degradation of RhB were performed using a UV1800 UV-Visible spectrophotometer. The initial dye concentration and catalyst content were 50mg/L and 2.0g/L, respectively. The dye solution was absorbed for 30min in the darkroom. Subsequently, the ultrasonic cleaning instrument with the frequency of 40kHz and power of 180W was turned on to generating ultrasonic vibration. After the elapse a small amount of dye solution was used for the examination of dye absorbance. The whole ultrasonic catalytic experiment was completed until 180min. The absorbance of dye at different time intervals was obtained by a UV1800 UV-Visible spectrophotometer. The degradation percentage of dye is defined as  $(A_0 - A_1)/A_0$ , where  $A_0$  and  $A_1$  are the absorbance of dye before and after irradiation, respectively.

## **3 RESULTS AND DISCUSSION**

#### 3.1 Phase Purity and Structure Analysis

The phase structure and purity of  $PbFe_{12}O_{19}$  magnetic separation catalysts with different Fe/Pb molar ratios and sintering temperatures are shown in Figure 2. Figure 2A shows the XRD patterns of  $PbFe_{12}O_{19}$  magnetic separation catalysts with a Fe/Pb molar ratio of 2. It was observed that all samples exhibited a mixed phase with low crystallinity including rhombohedral  $Fe_2O_3$ , PbO (£), or  $PbFe_4O_7$  ( $\nabla$ ). With the increase in sintering temperature, the diffraction peak intensity of the host lattice gradually increased, while the diffraction peaks of PbO disappeared. A similar phenomenon was observed

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when the Fe/Pb molar ratio was 3, as shown in Figure 2B. Figure 2C shows the XRD patterns of  $PbFe_{12}O_{19}$ magnetic separation catalysts with the Fe/Pb molar ratio of 4. The diffraction peak of PbFe<sub>4</sub>O<sub>7</sub> was absent when the sintering temperature increased from 600°C to 800°C. When the Fe/Pb molar ratio reached 5, the  $PbFe_4O_7$  impurity disappeared, as shown in Figure 2D. When the Fe/Pb molar ratio exceeded 6, a pure rhombohedral Fe<sub>2</sub>O<sub>3</sub> phase was observed above 600°C, as shown in Figure 2E-G. When the Fe/Pb molar ratio was 2, pure rhombohedral Fe<sub>2</sub>O<sub>3</sub> could be obtained by increasing the sintering temperature to 1100°C, as shown in Figure 2H. For the Fe/Pb molar ratio of 6, the sintering temperature above 600°C was pure rhombohedral Fe<sub>2</sub>O<sub>3</sub> phase. Comparing the samples of Fe/Pb molar ratios of 2, 3, 4, 5, 6, 9 and 12, only the samples for the Fe/Pb molar ratios of 2, 3 and 4 did not obtain pure rhombohedral  $Fe_2O_3$  at 800°C (Figure 2A-G), and only the samples for the Fe/Pb molar ratios of 2 did not obtain pure rhombohedral Fe<sub>2</sub>O<sub>3</sub> at 1100°C (Figure 2H). With the increase in temperature, the lead loss was more serious, which requires validation by XPS characterization. Figure 2I shows the XRD refinement result of Sample S7-800. All the diffracted lines for Sample S7-800 were consistent with the standard JCPDS card no. 33-0664 (rhombohedral  $Fe_2O_3$ ). The space group of rhombohedral Fe<sub>2</sub>O<sub>3</sub> was R-3c (167). The XRD diffraction peaks of Sample S7-800 were consistent with those of  $PbFe_{12}O_{19}$ reported in prior literature<sup>[25]</sup>.

Sözeri et al.<sup>[34]</sup> reported that the pure BaFe<sub>12</sub>O<sub>19</sub> particles could be obtained by the citrate sol-gel combustion method with an initial Fe/Ba molar ratio of 4. Rostami et al.<sup>[35]</sup> synthesized the pure  $PbFe_{12}O_{19}$ nanoparticles by the sol-gel method with the Fe/Pb molar ratio of 10. Thus, the influence of Fe/Pb molar ratio on the phase structure of PbFe<sub>12</sub>O<sub>19</sub> cannot be ignored. In the case of high Pb content, the excess Pb ions could not coordinate with Fe ions to form  $PbFe_{12}O_{19}$  but exist as PbO and PbFe<sub>4</sub>O<sub>7</sub>. During the process of high temperature sintering, the shortage of Pb in the system due to the volatilization of PbO gas in the reaction system causes the sample to show a phase that resembles Fe<sub>2</sub>O<sub>3</sub>. When the molar ratio of Fe/Pb reached 6 or above, Pb ions fully coordinated with Fe ions to form  $PbFe_{12}O_{19}$ . With the volatilization of PbO gas, a phase that resembles Fe<sub>2</sub>O<sub>3</sub> was obtained by high temperature treatment.

The crystallite size (D) of  $PbFe_{12}O_{19}$  magnetic separation catalysts was calculated by the Debye-Scherrer Equation (1) and the line broadening of the (104) diffraction peak.

$$D = \frac{k\lambda}{\beta\cos\theta} \quad (1)$$



Figure 1. Preparation flow chart of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts prepared by the polyacrylamide gel method with different Fe/Pb molar ratios.



**Figure 2. XRD patterns of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts.** A: Samples S1-600, S1-700 and S1-800; B: Samples S2-600, S2-700 and S2-800; C: Samples S3-600, S3-700 and S3-800; D: Samples S4-600, S4-700 and S4-800; E: Samples S5-600, S5-700 and S5-800; F: Samples S6-600, S6-700 and S6-800; G: Samples S7-600, S7-700 and S7-800; H: Samples S1-1100, S2-1100, S3-1100, S4-1100, S5-1100, S6-1100 and S7-1100; I: AXRD refinement result of Sample S7-800. The pink, blood red, blue curves and the vertical lines represent the observed value, the theoretically calculated values, the difference between the observed value and theoretically calculated value, and the Bragg peaks, respectively.

Where k is the shape factor, the k value is 0.62.  $\lambda$  is the X-ray wavelength and  $\theta$  is the Bragg diffraction angle.  $\beta$  is the full-width at half maximum (FWHM) on the basis of the pseudo-Voigt function<sup>[32]</sup>. The FWHM value of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts is a sum of the Gaussian and Lorentzian components.

Gaussian component:

$$(FWHM)_G = \sqrt{U \tan^2 \theta + V \tan \theta + W}$$
 (2)

Lorentzian component:

$$(FWHM)_L = X \tan \theta + \frac{Y}{\cos \theta}$$
 (3)

The interplanar spacing (d), the FWHM, and D values of  $PbFe_{12}O_{19}$  magnetic separation catalysts are given in Table 1. In Table 1, the D value of  $PbFe_{12}O_{19}$  magnetic separation catalysts increased with the increase in the sintering temperature. Some samples did not conform to this rule, which may be attributable to the interaction between lattice thermal expansion and coordination environment. Simultaneously, the D value of  $PbFe_{12}O_{19}$ magnetic separation catalysts was disordered with the increase of Fe/Pb molar ratio. For a more accurate analysis of these data, it is necessary to build a nonlinear dynamic model and use an artificial neural network algorithm model for further optimization and prediction.

#### **3.2 FTIR Analysis**

To further analyze the functional groups of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts, Figure 3 shows the FTIR spectra of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts. For all samples, two characteristic peaks at 3453 and 1638cm<sup>-1</sup> were observed in the wavenumber range of 4000-1500cm<sup>-1</sup>, which could be assigned to the stretching vibration of O-H bond and the bending vibration of H-O-H bond for the absorbed water, respectively<sup>[34,36,37]</sup>. The peak at 2365cm<sup>-1</sup> could be ascribed to the adsorbed carbon dioxide<sup>[37,38]</sup>. For the Samples S1 (Figure 3A), S2 (Figure 3B) and S3 (Figure 3C), two characteristic peaks at 1172 and 1122cm<sup>-</sup> <sup>1</sup> appear for samples sintered below 800°C, which can be attributed to C-O stretching vibrations<sup>[37,38]</sup>. When excessive Pb ions were introduced into the reaction system, the characteristic peaks of C-O stretching vibrations emerged at low sintering temperatures due to the presence of more carboxyl groups in Pb acetate, which need to be completely removed at higher sintering temperatures. At a low temperature, almost all PbFe<sub>12</sub>O<sub>19</sub> samples (Figure 3A-H) showed three characteristic peaks of 634, 533 and 458cm<sup>-1</sup>. The peaks at 533 and 458cm<sup>-1</sup> corresponded to the Fe-O and Pb-O bonds from  $PbFe_{12}O_{19}$  hexagonal ferrite, respectively<sup>[39-44]</sup>. The peak at 634cm<sup>-1</sup> was attributed to Pb-O bond from PbO or  $PbFe_4O_7^{[45]}$ . A characteristic peak at 442cm<sup>-1</sup> was also observed in Samples S1-700 and S2-700, which was mainly attributed to the Fe-O bond in  $PbFe_4O_7$ . When the sintering temperature increased, the characteristic peak disappeared, indicating that high temperature led to the loss of lead ions. Figure 3I shows the FTIR spectrum of Fe<sub>2</sub>O<sub>3</sub>. It could be seen from the figure that the two characteristic peaks in the low wavenumber segment were obviously different from those of  $PbFe_{12}O_{19}$ . The lead loss in  $PbFe_{12}O_{19}$  samples did not result in complete lead loss.

#### 3.3 XPS Analysis

The chemical compositions, surface electronic states, and electronic energy levels of  $PbFe_{12}O_{19}$  magnetic separation catalysts were studied by XPS, and the survey spectra, Pb 4d, Fe 2p, and O1s energy levels are shown in Figure 4. Figure 4A shows the survey spectra of  $PbFe_{12}O_{19}$  magnetic separation catalysts, revealing the presence of C, Pb, Fe and O elements. The result indicates that the Samples S5-800, S2-1100, S5-1100, and S6-1100 exhibit high crystallinity and contain no other impurity elements due to the C peak from the XPS instrument<sup>[46]</sup>. It could be seen from the figure that the characteristic peaks of Pb elements in Samples S2-1100, S5-1100, and S6-1100 were weaker than those in Sample S5-800, indicating that Pb loss emerged in  $PbFe_{12}O_{19}$ samples during high temperature sintering.

The Pb 4d spectra of  $PbFe_{12}O_{19}$  magnetic separation catalysts are presented in Figure 4B. Table 2 shows the XPS parameters of Samples S5-800, S2-1100, S5-1100, and S6-1100. As shown in Figure 4B, two characteristic peaks at 412.39-413.26 and 417.26-417.80eV were observed for the Pb 4d5/2 and Pb 4d3/2 from PbFe<sub>12</sub>O<sub>19</sub>, respectively. The characteristic peaks of Pb 4d in Samples S2-1100, S5-1100 and S6-1100 were characterized due to the obvious loss of lead in the hightemperature sintering process, which then results in an insignificant characteristic peak of Pb 4f.

The high-resolution Fe 2p spectra of  $PbFe_{12}O_{19}$ magnetic separation catalysts in Figure 4C show two major characteristic peaks at 710.15-710.84 and 723.64-724.87eV, with spin separation energy of 13.49eV, which is similar to a previous report of PbFe<sub>12</sub>O<sub>19</sub> and can be assigned to the Fe 2p3/2 and Fe 2p1/2, respectively<sup>[47,48]</sup>. Figure 4D shows the O 1s spectra of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts. The peak of O 1s spectra for the Samples S5-800 could be resolved into three components, indicating that three types of oxygen species were present on the surface of Samples S5-800, while the Samples S2-1100, S5-1100 and S6-1100 only exhibited two peaks. The peak at the binding energy of 531.96eV was typical for lattice oxygen of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts. The peak located at highest binding energy of 532.03-532.17eV could be assigned to adsorbed oxygen. The peaks at 529.69-530.29eV could be ascribed to the lattice

Table 1. The Interplanar Spacing	d), the Full-Width at Half Maximu	m (FWHM) and the C	<b>Crystallite Size</b>
of PbFe <sub>12</sub> O <sub>19</sub> Magnetic Separation	Catalysts		

Community.		Connetellite Circe (eV)			
Sample –	d (Å)	2θ (°)	FWHM (°)	- Crystallite Size (ev)	
Fe <sub>2</sub> O <sub>3</sub>	2.6994	33.160	0.181	45.285	
S1-600°C	2.6993	33.162	0.162	50.597	
S1-700°C	2.6993	33.162	0.175	46.838	
S1-800°C	2.6993	33.160	0.140	58.547	
S1-1100°C	2.6952	33.213	0.106	77.337	
S2-600°C	2.6993	33.161	0.188	43.599	
S2-700°C	2.6979	33.179	0.160	51.231	
S2-800°C	2.6978	33.180	0.150	54.647	
S2-900°C	2.6993	33.161	0.129	63.540	
S2-1000°C	2.6986	33.170	0.146	56.143	
S2-1100°C	2.6961	33.202	0.127	64.547	
S3-600 °C	2.7057	33.080	0.304	26.957	
S3-700°C	2.7010	33.139	0.182	45.034	
S3-800°C	2.7002	33.150	0.144	56.919	
S3-1100°C	2.6967	33.194	0.149	55.016	
S4-600°C	2.6964	33.198	0.265	30.934	
S4-700°C	2.7008	33.142	0.211	38.845	
S4-800°C	2.6986	33.170	0.126	65.054	
S4-1100°C	2.6963	33.161	0.125	65.573	
S5-600 °C	2.7009	33.141	0.167	49.079	
S5-700°C	2.7008	33.143	0.152	53.923	
S5-800°C	2.6961	33.201	0.171	47.939	
S5-900°C	2.6981	33.176	0.158	51.879	
S5-1000°C	2.6980	33.178	0.131	62.572	
S5-1100°C	2.6919	33.255	0.142	57.737	
S6-600°C	2.7010	33.139	0.191	42.912	
S6-700°C	2.7025	33.121	0.188	43.595	
S6-800°C	2.6933	33.238	0.168	48.799	
S6-1100°C	2.6976	33.183	0.134	61.172	
S7-600°C	2.7011	33.139	0.191	42.912	
S7-700°C	2.7010	33.139	0.200	40.981	
S7-800°C	2.6961	33.202	0.159	51.557	
S7-1100°C	2.6977	33.182	0.136	60.273	

oxygen of Fe<sub>2</sub>O<sub>3</sub>. Thus, PbFe<sub>12</sub>O<sub>19</sub> could also be written as (PbO)·6(Fe<sub>2</sub>O<sub>3</sub>). When the sintering temperature further increased, the energy levels of lattice oxygen and adsorbed oxygen shifted to higher energy levels. The characteristic peak of PbFe<sub>12</sub>O<sub>19</sub> lattice oxygen in Sample S5-800 was obvious, but the characteristic peak of PbFe<sub>12</sub>O<sub>19</sub> lattice oxygen in Samples S2-1100, S5-1100 and S6-1100 disappeared. The results indicate that the lead loss causes PbFe<sub>12</sub>O<sub>19</sub> to phase transition to Fe<sub>2</sub>O<sub>3</sub> and PbO<sup>↑</sup>.

### 3.4 Surface Morphology Analysis

With the increase in sintering temperature, the particle

size of  $PbFe_{12}O_{19}$  increased, and the agglomeration between particles became more and more obvious. Figure 5A-G shows the SEM images of Samples S1-1100, S2-1100, S3-1100, S4-1100, S5-1100, S6-1100 and S7-1100. When the sintering temperature rose to 1100°C, the adhesion agglomeration phenomenon between  $PbFe_{12}O_{19}$  particles with Fe/Pb mole ratio of 2 was more obvious, and the interaction between particles grew into irregular lotus root particles, as shown in Figure 5A. However, the samples with Fb/Pb mole ratio of 3 showed human bone-like particles, as shown in Figure 5B. The ingot-shaped particles were bonded into flake particles



Figure 3. FTIR spectra of  $PbFe_{12}O_{19}$  magnetic separation catalysts. A: Samples S1-600, S1-700 and S1-800; B: Samples S2-600, S2-700 and S2-800; C: Samples S3-600, S3-700 and S3-800; D: Samples S4-600, S4-700 and S4-800; E: Samples S5-600, S5-700 and S5-800; F: Samples S6-600, S6-700 and S6-800; G: Samples S7-600, S7-700 and S7-800; H: Samples S1-1100, S2-1100, S3-1100, S4-1100, S5-1100, S6-1100 and S7-1100; I:  $Fe_2O_3$ .



Figure 4. XPS spectra of Samples S5-800, S2-1100, S5-1100 and S6-1100. A: Survey spectra; B: Pb 4d energy level; C: Fe 2p energy level; D: O1s energy level.

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Table 2. XPS Parameters of Sam	ples S5-800,	S2-1100, S5-2	1100 and S6-1100
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Samples —	Pb4d (eV)		Fe2p (eV)				
	Pb4d5/2	Pb4d3/2	Fe2p3/2	Fe2p1/2			
S5-800	412.39	417.26	710.32	723.64	532.15	531.96	529.96
S2-1100	412.75	417.57	710.54	724.26	532.08	/	530.06
S5-1100	413.26	417.74	710.84	724.87	532.17	/	530.29
S6-1100	412.83	417.80	710.15	724.09	532.03	/	529.69



Figure 5. SEM images of Samples. A: S1-1100; B: S2-1100; C: S3-1100; D: S4-1100; E: S5-1100; F: S6-1100; G: S7-1100; H: TEM; I: High-resolution transmission electron microscope images of Sample S7-1100.

mixed with fine lotus root-shaped particles, as shown in Figure 5C. The rice-like grains grew further and became nearly spherical particles, as shown in Figure 5D. Figure 5E shows the SEM image of S5-1100. The rhomboid particles and fine particles were further grown and bonded into dense blocks with very clear grain boundaries between the particles. In Figure 5F, the uniform rhomboid particles grew to cohesive and agglomerated spherical particles, which may be related to the loss of lead ions. In Figure 5G, the homogeneous lamellar PbFe<sub>12</sub>O<sub>19</sub> particles obtained at 1100°C also grew into dense spherical particles with a small number of lamellar particles. Figure 5H shows the TEM image of Sample S7-1100. As can be seen from Figure 5H, adhesion agglomeration between

particles was particularly evident, and the particles grew to cross-link with each other. Figure 5I shows the highresolution transmission electron microscope image of Sample S7-1100. The result shows that the lattice spacing of 0.2514nm corresponded to the d-spacing of (110) planes, and 0.2671nm corresponded to (104) planes. The grain growth for the PbFe<sub>12</sub>O<sub>19</sub> particles was dominated by (110) and (104) crystal planes. The surface morphology of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts could be effectively adjusted by adjusting Fe/Pb molar ratio and sintering temperature.

## **3.5 Optical Properties**

To investigate the effects of Fe/Pb molar ratio and



Figure 6. UV-Vis diffuse reflection spectra of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts. A: Samples S1-600, S1-700 and S1-800; B: Samples S2-600, S2-700 and S2-800; C: Samples S3-600, S3-700 and S3-800; D: Samples S4-600, S4-700 and S4-800; E: Samples S5-600, S5-700 and S5-800; F: Samples S6-600, S6-700 and S6-800; G: Samples S7-600, S7-700 and S7-800; H: Samples S1-1100, S2-1100, S3-1100, S4-1100, S5-1100, S6-1100 and S7-1100; I: Fe<sub>2</sub>O<sub>3</sub>.

sintering temperature on the color properties, optical absorption coefficients and optical band-gap (Eg) values of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts, Figure 6 shows the UV-Vis diffuse reflection spectra of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts. Based on the data on color properties, it can be further confirmed that the PbFe<sub>12</sub>O<sub>19</sub>, which resembles the  $Fe_2O_3$  phase, is not  $Fe_2O_3$ . With an increase in the wavelength from 190 to 1100nm, the reflectivity altered obviously. In the range of 190-550nm, the reflectivity remained constant and hardly varied with wavelength. In the range of 550-750nm and 900-110nm, the reflectivity increased sharply with the increase in wavelength. With the increase in sintering temperature and Fe/Pb mole ratio, the reflectance trend was disordered. The color coordinated (L\* (black (0)/white (100)), a\* (green (-)/red (+)), b\* (blue (-)/yellow (+))), the chroma parameter (c\*), the hue angle (H°) and the total color difference (DE<sub>CIE</sub>\*) of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts were calculated on the basis of UV-visible diffuse reflectance spectrum and previous reports<sup>[49]</sup>.

The (L\*, a\*, b\*), c\*, H°, and  $DE_{CIE}$ \* values of

PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts are shown in Table 3. There was no linear relationship between (L\*, a\*, b\*), c\*, H° and DE<sub>CIE</sub>\* values of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts and sintering temperature or Fe/Pb molar ratio. This kind of nonlinear relation is especially suitable for the study of artificial neural network model. It is noteworthy that the b\* values of all samples obtained by sintering at 1100°C are negative, indicating the presence of blue components in these samples, which is consistent with the real photos in Figure 7. The color of  $Fe_2O_3$  is light red, and the presence of this blue component indicates that the phase structure that resembles Fe<sub>2</sub>O<sub>3</sub> should be ascribed to PbFe<sub>12</sub>O<sub>19</sub>. Based on the Kubelka-Munk and Tauc relationship (Equations (4) and (5)), the Eg values of  $PbFe_{12}O_{19}$  magnetic separation catalysts are given in Table 3. Similarly, Eg value is linearly independent of sintering temperature or Fe/Pb molar ratio.

$$F(R) = \frac{\alpha}{S} = \frac{(1 - R_{\infty})^2}{2R}$$
 (4)

Where R is the reflectance,  $\alpha$  is the optical absorption



Figure 7. Real photos of Samples S1, S2, S3, S4, S5, S6 and S7 sintered at different temperatures.

coefficient and S is the scattering coefficient.

 $(\alpha h v)^n = A(h v - E_g) (5)$ 

Where hn is the photon energy and A is a constant.

# **3.6 Magnetic Properties and Ultrasonic Catalytic Activity**

### **3.6.1 Magnetic Properties**

Figure 8A shows the hysteresis loops of Samples S7-600, S7-700, S7-800, and S7-1100, and the saturation magnetization of Samples S7-600, S7-700, S7-800 and S7-1100 are 0.53, 0.65, 0.76 and 0.72emu/g, respectively. The remaining magnetization of Samples S7-600, S7-700, S7-800 and S7-1100 were 0.23, 0.28, 0.53 and 0.45emu/g, respectively. The squareness ratio (remaining magnetization/saturation magnetization) of Samples S7-600, S7-700, S7-800 and S7-1100 were 0.43, 0.43, 0.70 and 0.63, respectively. The intrinsic coercivity (Hc) of Samples S7-600, S7-700, S7-800 and S7-1100 were

2752.43, 2882.73, 1742.67 and 1221.49Oe, respectively. This result confirms that the Samples S7-600, S7-700, S7-800 and S7-1100 exhibit the isotropic characteristic, so  $PbFe_{12}O_{19}$  can be applied as a promising material for the magnetic separation catalysts.

# **3.6.2 Effect of Different Sintering Temperatures on the Ultrasonic Catalytic Activity**

Figure 8B shows the degradation percentage of Samples S5-600, S5-700, S5-800, S5-900, S5-1000, and S5-1100. With the increase in sintering temperature, the ultrasonic catalytic activity of  $PbFe_{12}O_{19}$  magnetic separation catalyst decreased first, increased, then decreased. There are two main reasons: first, the phase structure of the sample sintered at above 600°C changed significantly, and the other is that the coordination environment of  $PbFe_{12}O_{19}$  changes due to the loss of some lead ions during high temperature treatment, which affects the surface oxygen vacancy concentration of  $PbFe_{12}O_{19}$ .

# Table 3. Color Correlation Parameters and Eg Values of PbFe<sub>12</sub>O<sub>19</sub> Magnetic Separation Catalysts

		Color Coordinates					
Sample –	L*	a*	b*	<b>c</b> *	H°	DE <sub>CIE</sub> *	– Eg Value (eV)
Fe <sub>2</sub> O <sub>3</sub>	31.319	18.522	9.015	20.599	25.953	37.486	1.97
S1-500°C	39.194	18.590	12.522	22.414	33.964	45.150	1.96
S1-600°C	33.698	10.675	3.700	11.298	19.117	35.542	1.81
S1-700°C	32.061	13.212	6.520	14.733	26.266	35.284	1.84
S1-800°C	29.921	9.140	3.166	9.673	19.106	31.446	1.73
S1-1100°C	31.168	2.520	-2.215	3.355	-41.314	31.348	1.34
S2-500°C	42.570	19.125	14.520	24.012	37.206	48.875	1.94
S2-600°C	36.119	12.858	6.111	14.236	25.420	38.823	1.83
S2-700°C	33.544	10.310	3.419	10.862	18.347	35.259	1.75
S2-800°C	38.256	11.561	4.752	12.450	22.344	40.246	1.87
S2-900°C	35.341	11.522	4.548	12.387	21.540	37.449	1.84
S2-1000°C	32.332	8.913	1.778	9.089	11.281	33.585	1.77
S2-1100°C	32.516	0.148	-3.536	3.539	-87.603	32.708	1.14
S3-500°C	45.938	18.181	15.778	24.073	40.952	51.863	1.98
S3-600°C	45.801	16.061	11.188	19.574	34.861	49.808	1.94
S3-700°C	37.830	10.233	3.361	10.771	18.183	39.333	1.78
S3-800°C	34.367	9.472	3.415	10.069	19.826	35.812	1.82
S3-1100°C	32.764	0.858	-3.170	3.284	-74.855	32.928	1.19
S4-500°C	45.657	18.556	16.029	24.520	40.821	51.825	1.95
S4-600°C	40.606	12.324	4.738	13.203	21.029	42.699	1.86
S4-700°C	43.106	13.770	7.282	15.577	27.871	45.834	1.90
S4-800°C	32.943	7.831	1.487	7.971	10.752	33.894	1.78
S4-1100°C	34.157	0.773	-3.416	3.502	-77.249	34.336	1.23
S5-500°C	42.942	16.372	10.886	19.661	33.621	47.229	1.93
S5-600°C	34.502	9.027	2.009	9.248	12.547	35.720	1.77
S5-700°C	33.845	7.128	0.969	7.194	7.741	34.601	1.69
S5-800°C	37.234	8.818	2.891	9.280	18.152	38.373	1.76
S5-900°C	39.710	11.212	3.669	11.797	18.120	41.425	1.89
S5-1000°C	37.009	8.088	1.021	8.152	7.195	37.896	1.78
S5-1100°C	35.469	3.821	-2.013	4.319	-27.781	35.731	1.39
S6-500°C	36.839	18.491	12.113	22.105	33.228	42.962	1.94
S6-600°C	38.661	12.077	3.659	12.619	16.855	40.668	1.89
S6-700°C	40.085	10.671	3.327	11.178	17.316	41.614	1.86
S6-800°C	37.089	12.992	6.134	14.367	25.274	39.775	1.88
S6-1100°C	34.681	3.059	-2.411	3.895	-38.244	34.899	1.36
S7-500 °C	41.417	21.707	15.950	26.937	36.308	49.406	1.99
S7-600°C	37.916	11.941	3.111	12.340	14.603	39.873	1.89
S7-700°C	39.962	13.314	5.685	14.477	23.122	42.503	1.93
S7-800°C	37.335	11.803	4.347	12.578	20.219	39.397	1.86
S7-1100°C	35.410	3.296	-2.265	3.999	-34.497	35.635	1.38

Notes: L\*, a\*, b\*: the color coordinates; c\*: the chroma parameter;  $H^\circ$ : the hue angle;  $DE_{CIE}$ \*: the total color difference.

# 3.6.3 Effect of Different Fe/Pb Mass Ratios on the Ultrasonic Catalytic Activity

Figure 8C-F shows the degradation percentage of different samples obtained at 600, 700, 800, and 1100°C.

When Samples S1-S7 were sintered at 600 (Figure 8) and 700°C (Figure 8D), the degradation percentage increased first and then decreased with the increase in Fe/Pb mole ratio, and Sample S5 showed the best

(i)



Figure 8. Hysteresis loops of Samples S7-600, S7-700, S7-800, and S7-1100 (A), degradation percentage of Sample S5 obtained at different temperatures (B), and degradation percentage of different samples obtained at (C) 600, (D) 700, (E) 800 and (F) 1100°C.

degradation percentage. When the sintering temperature reaches 800°C (Figure 8E), the degradation percentage of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts increased first, decreased, and increased with the increase in the mole ratio of Fe/Pb, and reached the maximum value when the mass ratio of Fe/Pb was 12. When the sintering temperature reached 1100°C (Figure 8F), the degradation percentage decreased with the increase of Fe/Pb mole ratio, and the Sample S7 appeared abnormal. The degradation percentage varied significantly with the mass ratio of Fe/Pb at different sintering temperatures due to the loss of Pb and the change in the structure and composition of PbFe<sub>12</sub>O<sub>19</sub>.

# 3.7 Multi-parameter Coupling Artificial Neural Network Algorithm Simulation

An artificial neural network algorithm is an efficient method to predict the physical and chemical properties of semiconductor materials<sup>[50,51]</sup>. On the basis of obtaining a large number of data, the artificial neural network model was established to train the data to conform to the corresponding rules. The known input parameters were then used as the test set to predict the output results. In this experiment, various parameters including d, 20, FWHM, D, degradation percentage, Eg, L\*, a\*, b\*, c\*, H° and  $DE_{CIE}^{*}$  values were obtained by controlling Fe/Pb mole ratio, the mass of Fe and Pb metal salts and sintering temperature. By analyzing the necessity of the experiment, some experimental data were not obtained. To study the influence of different parameters on the ultrasonic catalytic activity of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts, three different artificial neural network models were established to predict the related parameters. Artificial neural network algorithm model I (Figure 9) was established by using the Fe/Pb molar ratio, the mass of Fe and Pb metal salts, sintering temperature, d, 20, FWHM, L\*, a\*, c\*,  $H^{\circ}$  and  $DE_{CE}^{*}$  values as the input parameters, and the most important parameters including D, degradation percentage, Eg and b\* values as the output parameters. The effects of different parameters on the D, degradation percentage, Eg and b\* values of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts were observed. The R<sup>2</sup> value was close to 1, and the predicted value was highly consistent with the experimental value, as shown in Figure 10. Figure 9B shows the artificial neural network algorithm model II for prediction of degradation percentage for different samples. This model mainly uses all parameters except degradation percentage as input parameters. The synergistic effect of multiple parameters has a great influence on the degradation percentage, and the  $R^2$ value is closer to 1 than model I. Figure 11A shows the experimental values of the degradation rate of different samples by Matlab software and the simulated values by the artificial neural network algorithm. The experimental values of the seven test samples were in good agreement with the predicted values. To predict the physical and chemical properties of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts with only the experimental parameters, an artificial neural network model III was designed. Figure 9C shows the artificial neural network algorithm model III for prediction of d,  $2\theta$ , FWHM, D, degradation percentage, Eg, L\*, a\*, b\*, c\*, H° and DE<sub>CIE</sub>\* values for different samples. With only four input parameters of Fe/Pb mole ratio, the mass of Fe and Pb metal salts and sintering temperature, the d, 20, FWHM, D, degradation



**Figure 9.** Artificial neural network algorithm models. A: Artificial neural network algorithm model I for prediction of D, degradation percentage, Eg and b\* values for different samples; B: Artificial neural network algorithm model II for prediction of degradation percentage for different samples; C: Artificial neural network algorithm model III for prediction of d, 20, FWHM, D, degradation percentage, Eg, L\*, a\*, b\*, c\*, H° and  $\Delta E_{CIE}$ \* values for different samples.

percentage, Eg, L\*, a\*, b\*, c\*, H° and  $DE_{CIE}^{*}$  values of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts were predicted. The experimental value and the simulated value of artificial neural network algorithm of d, 2 $\theta$ , FWHM, D, degradation percentage, Eg, L\*, a\*, b\*, c\*, H° and DE<sub>CIE</sub>\* values for different samples by Matlab software are shown in Figure 11B. Based on artificial neural network model III, other physical and chemical properties of  $PbFe_{12}O_{19}$  magnetic separation catalysts not obtained in this experiment can be predicted.

#### **4 CONCLUSION**

The  $PbFe_{12}O_{19}$  magnetic separation catalysts with different Fe/Pb molar ratios were synthesized by a simple polyacrylamide gel method. XRD, FTIR, and XPS characterizations confirmed that the crystal structure and



Figure 10. The experimental value and the simulated value of artificial neural network algorithm of (A) D, (B) degradation percentage, (C) Eg and (D) b\* values for different samples by Matlab software.



**Figure 11. The experimental value and the simulated value of artificial neural network algorithms.** A: The experimental value and the simulated value of artificial neural network algorithm of degradation percentage for different samples by Matlab software; B: The experimental value and the simulated value of artificial neural network algorithm of d, 20, FWHM, D, degradation percentage, Eg, L\*, a\*, b\*, c\*, H° and  $\Delta E_{CIE}^*$  values for different samples by Matlab software.

composition of  $PbFe_{12}O_{19}$  magnetic separation catalysts altered with the change in Fe/Pb mole ratio and sintering temperature. The surface morphology of  $PbFe_{12}O_{19}$ magnetic separation catalysts could be regulated and controlled by different Fe/Pb molar ratios and sintering temperatures. The color relative parameters and Eg values of  $PbFe_{12}O_{19}$  magnetic separation catalysts were linearly independent of Fe/Pb mole ratio and sintering temperature. The  $PbFe_{12}O_{19}$  magnetic separation catalysts exhibiting a ferromagnetic behavior indicate that the catalysts could be separated through a magnetic field. The maximum degradation percentage of 78.99% for the  $PbFe_{12}O_{19}$ magnetic separation catalysts with Fe/Pb mole ratio of 12 was found at 2g/L of initial catalyst concentration, 50mg/L of initial dye concentration, 180min of vibration time for removal of RhB dye from wastewater. Based on the experimental parameters, including Fe/Pb molar ratio, sintering temperature, the mass of lead and iron metal salts, and analysis results including the interplanar spacing, FWHM, D, colorimetric parameter, Eg value and degradation percentage of the target product, a neural network algorithm model was established to predict the effects of different experimental parameters on the D, colorimetric parameter, Eg value and degradation percentage of PbFe<sub>12</sub>O<sub>19</sub> magnetic separation catalysts. The application of neural network algorithm to predict the physical and chemical properties of other semiconductor materials provides theoretical guidance. Simultaneously, this technology provides experimental basis for the development of novel magnetic separation ultrasonic catalysts.

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# **Conflicts of Interest**

The authors declared that they have no competing interests.

## **Author Contribution**

Chen X: Experiment, data analysis and paper writing; Liu H and Li M: Performance prediction; Wang S: Supervision and paper revision.

# **Abbreviation List**

D, Crystallite size Eg, Optical band-gap FTIR, Fourier transform infrared FWHM, Full-width at half maximum PbFe<sub>12</sub>O<sub>19</sub>, Hexagonal lead ferrite RhB, Rhodamine B SEM, Scanning electron microscopy TEM, Transmission electron microscopy UV-Visible, Ultraviolet-visible XPS, X-ray photoelectron spectrometer XRD, X-ray diffractometer

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